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### NATIONAL ADVISORY COMMITTEE FOR AERONAUTICS

### TECHNICAL NOTE 3017

AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND UNNOTCHED SHEET

SPECIMENS OF 618-T6 ALUMINUM ALLOY, ANNEALED

347 STAINLESS STEEL, AND HEAT-TREATED

403 STAINLESS STEEL

By Herbert F. Hardrath, Charles B. Landers, and Elmer C. Utley, Jr.

Langley Aeronautical Laboratory
Langley Field, Va.



Washington

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### SUMMARY

Axial-load fatigue tests at a stress ratio of zero were performed on notched and unnotched sheet specimens of 615-T6 aluminum alloy and 347 and 403 stainless steels. Special emphasis was placed on tests at high stress levels which produce failures in small numbers of cycles. The stress-concentration factors effective in fatigue of notched specimens were found to be somewhat less than the theoretical elastic values at low stresses and were approximately equal to one at the ultimate strength. The minimum life to failure at stresses near the ultimate strength was drastically reduced with increasing stress-concentration factor.

### INTRODUCTION

The experimental investigation reported herein was carried out to provide information on the fatigue properties of 61S-T6 aluminum alloy, annealed 347 stainless steel, and heat-treated 403 stainless steel. Unnotched specimens and specimens containing notches were tested under repeated tensile stresses at a stress ratio (ratio of minimum stress to maximum stress) of zero. Since a knowledge of fatigue properties at high stresses is useful in some design problems, this investigation included tests in this range.

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The primary purpose of this paper is to present the results of the tests. Some comparisons with other work are also included.

The materials used to prepare specimens for this investigation were supplied by Bell Aircraft Company.

### SYMBOLS

KF	stress-concentration factor effective in fatigue (ratio of stress in an unnotched specimen at a given lifetime to stress in a notched specimen at same lifetime)
К <sub>Р</sub>	plastic stress-concentration factor (ratio of maximum local plastic stress to average stress in net section)
KT	theoretical stress-concentration factor (ratio of maximum local elastic stress to average stress in net section)
N	cycles
R	stress ratio (ratio of minimum stress to maximum stress)
S	average stress in net section

### MATERIALS

The 61S-T6 aluminum-alloy material used in the present tests came from a single sheet 4 feet wide, 12 feet long, and 0.125 inch thick. The sheet was painted with zinc chromate to protect the surface during specimen preparation. The sheet was cut into blanks according to the layout shown in figure 1 and each blank was labeled as indicated. Standard specimens (ref. 1) for tensile and compressive static tests were cut from blanks taken at random from the sheet.

The annealed 347 stainless-steel material also came from a single sheet 3 feet wide, 10 feet long, and 0.004 inch thick, painted with zinc chromate. The specimen blanks were cut and labeled as indicated in figure 2.

The 403 stainless-steel material was cut from two sheets 3 feet wide, 10 feet long, and 0.050 inch thick. These sheets were cut into pieces  $7\frac{3}{4}$  by 17 or  $10\frac{1}{4}$  by 17 as indicated by the heavy lines in figure 2 and were heat-treated to Rockwell C 40 to 41 by Bell Aircraft Company. The location of these pieces within the original sheets was not available. The pieces were, therefore, arbitrarily numbered in consecutive order and each piece was cut into specimen blanks, as shown by the light lines in figure 2, to provide notched and unnotched specimens. Since the material had become warped during heat treatment, specimens were machined from blanks which were selected for minimum

warpage or with warped portions remote from the central portion of the specimen. The effects of the warpage on the results is discussed subsequently.

### SPECIMEN PREPARATION

The dimensions of specimens used in this investigation are given in figure 3. The notched specimens are similar to those tested at Battelle Memorial Institute (ref. 2) and have elastic stress-concentration factors  $K_{\rm T}$  equal to 2 and 4 (ref. 3).

As is known, the technique used in specimen preparation can have an important effect on the results of fatigue tests (see ref. 4). The following explanation of the procedures used in preparing specimens for this investigation is therefore given in detail. In general, these procedures are felt to have produced little residual stress in the machined surfaces, but no detailed studies were made to obtain a quantitative check.

The unnotched specimens were clamped in stacks about 1 inch thick and machined in a lathe to produce the 12-inch radius of curvature at the edges. Successively lighter cuts were taken with the last two or three cuts removing about 0.0005 inch. The material was rotated at a speed of approximately 30 rpm.

The notched specimens were machined along the parallel edges in stacks and then the notches were machined in each specimen separately. The specimen was mounted on a combination turn-table and cross-slide support and the notches were cut with a milling cutter rotating about an axis normal to the plane of the sheet. Milling tools with helical cutting edges and 5/16-inch diameters were used to cut the specimens which have a notch radius of 0.3175 inch. The cutter speed was constant at 1,500 rpm for 61S-T6 aluminum-alloy specimens and at 675 rpm for stainless-steel specimens. Very slow manual feeds were used. Each cut removed 0.0005 inch or less in the final stages of machining. The same procedure was used for notches with a radius of 0.057 inch except that cutters with 0.100-inch diameter were used.

The surfaces of all specimens were left unpolished, but sharp edges were slightly rounded by hand with fine emery paper. The paper was moved in a longitudinal direction to leave no transverse scratches. In the case of the 61S-T6 notched specimens, the edges in the notches were removed with a pad of steel wool spinning in the jaws of a 1/4-inch drill. The specimen was held against this pad with very light pressure so that only the edges were cut. The sharp edges of notches in stainless-steel specimens were removed with emery paper rolled into a small cylinder and rotated by hand.

All specimens were tested under axial load at a stress ratio of zero. Three types of testing machines were used to cover the complete range of the S-N curves.

Most of the tests were performed in subresonant fatigue testing machines with capacities of 20,000 pounds. These machines and the associated load measuring apparatus are described in detail in reference 5. The probable error of the load measuring apparatus is approximately 1 percent. Frequent monitoring revealed that the loads rarely change ind much as 5 percent during any given test.

Tests in which failure occurred in less than 10,000 cycles were impractical to perform with these fatigue testing machines because of the trial-and-error procedure required to start each test. Consequently, a machine hydraulically operated at 100 cpm was used for tests in which failure was expected to occur in 500 to 10,000 cycles. Tests in which failure was expected to occur in less than 500 cycles were performed in static testing machines which were manually controlled to apply loads at approximately of cpm.

All specimens except those tested in static testing machines were clamped within rule plates similar to those described previously (refs. and a). For all specimens tested at stress levels higher than the yield strength of the material, the first cycle of load was applied manually to produce the plastic deformation corresponding to that load. This procedure simplified the maintenance of the desired mean load at the start of each of these tests.

### TEST RESULTS AND DISCUSSION

### Static Tests

The results of static tensile and compressive tests on standard test coupons are presented in table I. The 61S-Tó aluminum-alloy material had properties exceeding the minimum mechanical properties listed in table 3.111(f) of reference. The 34° stainless-steel material had properties exceeding those listed in table 2.111(c) of reference. Stress-strain curves for each material were obtained by averaging four autographically recorded curves and are presented in figure 4.

### Static Tests of Fatigue Specimens

Results of static tensile tests of each type of fatigue specimen are included in tables II to IV and figures 5 to 7. The differences between static strengths of notched and unnotched specimens made of the same material appear to be outside the range of probable error in the tests. In all but one case (347 stainless-steel material with Kp = 4) the notched specimens had greater static strengths than the unnotched specimens made of the same material. The increased strength in notched specimens can be considered to be due to the development of a multiaxial tensile stress which, in effect, reduces the maximum shear stress and thus retards fracture; however, the present knowledge of static strength of notched parts does not permit quantitative predictions.

### Fatigue Tests

The results of fatigue tests on 61S-Tú aluminum-alloy specimens are given in table II and are plotted in the form of S-N curves in figure 5. Similarly, the data for 34% and 403 stainless-steel specimens are given in tables III and IV and are plotted in figures 6 and 7, respectively. In the presentation of data, no distinction is made among unnotched specimens failing within the middle inch of the specimen. In this region the stresses are within 3 percent of the stress which occurs at the minimum section. Unnotched specimens occasionally failed at sections somewhat more removed from the middle of the specimen. Since stresses at these sections were definitely out of the range of possible variations in applied stresses, the specimens which failed outside the middle inch are identified in the tables and figures.

In the warped 403 stainless-steel specimens the minimum radius of curvature of the sheet surface, as measured by a curvature gage with a 5-inch gage length, was 50 inches; thus, the maximum stress resulting from clamping the specimens between flat plates was approximately 15 ksi. This stress, however, usually occurred in sections remote from the central portion of the specimens. Where a curvature appeared at the critical section, only 30 percent of the specimens developed initial fatigue cracks on the side of the specimen where the bending stress was tensile. These observations lead to the conclusion that the stresses due to bending did not affect the results significantly.

### Minimum Life at High Stresses

In tests on unnotched specimens those which survived 1 cycle of load near the ultimate tensile strength were found to withstand approximately 10<sup>4</sup> cycles of that load before failure occurred. The S-N curves (figs. 4 to 6) are, therefore, shown dashed between this minimum life

and 1/2 cycle. An exception appears in the tests of the 347 stainless-steel material where four failures occurred between 100 and 10<sup>3</sup> cycles. This observation of minimum life to failure in unnotched specimens appears to be consistent with most tests previously reported.

For specimens containing notches with an elastic stress-concentration factor  $K_T$  of 2, those surviving 1 cycle of load usually survived approximately  $10^3$  cycles before failure. Similarly, specimens containing notches with  $K_T=4$  generally survived 100 cycles before failure if they survived the first cycle. Thus, the minimum life was reduced by a factor of 10 each time the theoretical stress-concentration factor was doubled. No way of predicting this behavior is apparent at present. The same ratios of cycles to failure are found to exist for stresses in the vicinity of two-thirds of the ultimate strength.

### Fatigue Stress-Concentration Factors

At lower stresses the S-N curves for unnotched specimens and specimens containing notches appear to be roughly parallel. Fatigue stress-concentration factors  $K_{\Gamma}$  have been computed by dividing the stress in an unnotched specimen by the nominal stress in a notched specimen which failed in the same number of cycles. These factors are plotted against maximum average stress in the net sections in figures 8, 9, and 10 for the three materials. In each case the curve for  $K_{\Gamma}$  is seen to have a maximum value less than  $K_{\Gamma}$  for low stresses and progressively lower values for higher stresses. The differences between  $K_{\Gamma}$  and the maximum value of  $K_{\Gamma}$  may be expected to be due to size effect (ref. 8). Since, however, current predictions for size effect in fatigue are restricted to completely reversed stress cases, no prediction of the magnitude of the maximum value of  $K_{\Gamma}$  is possible at this time.

Previous work (refs. 3, 9, and 10) has shown that plastic deformation reduces the severity of stress concentration during the first load application and that the magnitude of the plastic stress-concentration factors Kp can be predicted as long as the strains are small. Griffith (ref. 9) has also shown that, at least during the first 100 cycles of a load which produces plastic deformations at a discontinuity, the local strains and stresses oscillate between the values observed at the end of the first 1/2 cycle and at the end of the unloading part of the first cycle.

If the local stresses in notched specimens are assumed to remain unchanged throughout a fatigue test and are assumed to produce fatigue failures in the same number of cycles as do stresses of the same magnitude

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in unnotched specimens, then the curves of K<sub>F</sub> and K<sub>P</sub> would be expected to be equivalent. The dashed curves in figures 8 to 10 represent the curves for K<sub>P</sub> which were calculated for each of the notched specimens by use of the stress-strain curves appropriate for each material and the method described in reference 3.

For the 61S-T6 aluminum-alloy material (fig. 8) the curve of KF lies below the curve for Kp in the region of stresses less than about 15 ksi for  $K_T = 4$  and 28 ksi for the  $K_T = 2$  configurations. These stress levels correspond to approximately  $7 \times 10^4$  cycles to failure (see fig. 4) and this life is approximately the minimum life of unnotched specimens which failed under repeated stress. For higher stresses the curves of  $K_T$  and  $K_T$  are in fair agreement. The agreement at high stresses, however, does not permit the use of the curve of  $K_T$  in the prediction of the S-N curves for notched specimens inasmuch as it occurs in a region where the S-N curves for unnotched specimens are horizontal.

In the case of the 347 stainless-steel material (fig. 9) the curves of K<sub>F</sub> lie above the curves for K<sub>P</sub> for all stress levels. This lack of agreement is probably due to the fact that the endurance limit for unnotched specimens (55 ksi) is approximately 20 percent higher than the yield strength (45.6 ksi) and corresponds to approximately  $2\frac{1}{2}$  percent strain on the tensile stress-strain curve for the material. Since such large plastic deformations are experienced before any failure occurs by fatigue, a relation between fatigue stress-concentration factors and the original stress-strain properties of the material probably cannot be expected.

The comparison between Kr and Kp for the 403 stainless-steel material (fig. 10) is similar to that for the 61S-T6 aluminum-alloy material as discussed previously.

### CONCLUDING REMARKS

Sheet specimens containing no notches or notches with theoretical stress-concentration factors  $K_T$  of 2 and 4 and made of 61S-T6 aluminum alloy, annealed 347 stainless steel, and heat-treated 403 stainless steel have been tested under axial load at a stress ratio of zero. The results indicate that failures in unnotched specimens subjected to repeated stresses near the ultimate strength occurred in roughly  $10^4$  cycles; in notched specimens with  $K_T=2$  failure occurred in about  $10^3$  cycles; and in notched specimens with  $K_T=4$  failure occurred in about 100 cycles.

In the range of stress where failure occurred by fatigue, the effective stress-concentration factor  $K_{\rm F}$  decreased from a maximum value somewhat less than the theoretical factor at low stresses to a minimum value approaching or less than one at the static failing stress. A comparison between  $K_{\rm F}$  and plastic stress-concentration factors  $K_{\rm P}$  revealed that in the cases of the 61S-T6 aluminum-alloy material and the 403 stainless-steel material the values were approximately the same at high stresses but in the case of the 347 stainless-steel material there appears to be no correlation between the two factors.

Langley Aeronautical Laboratory,
National Advisory Committee for Aeronautics,
Langley Field, Va., June 23, 1953.

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TABLE I. - TENSILE AND COMPRESSIVE PROPERTIES OF MATERIALS TESTED

Material	Ultimate tensile strength, ksi	Tensile yield strength (offset = 0.2 percent), ksi	Elongation in 2 inches, percent	Compressive yield strength (offset = 0.2 percent), ksi
618-T6 aluminum alloy	17.0 842.0	<sup>μ</sup> 2.0 836.0	17 <b>5</b> 10	42.8 a35.0
347 stainless steel	92.0 c75.0	45.6 530.0	61	29.9
403 stainless steel	190.0	153.0	8	160.8

CValues obtained from table 2.111(c) in ref. 7. Walues obtained from table 5.111(f) in ref. 7. byalue obtained from table 38 in ref. 11.

Table II.- Fatigue test results for 61s-t6 aluminum-alloy material under direct stress at R=0

### (a) Unnotched sheet specimens

Specimen	Maximum stress, ksi	Fatigue life, cycles	Speed,	Remarks
D1B10 D1B4 D1B45 D1B46 D1B39 D1B16	47 47 46.4 46.3 46.2 46	38,000 56,000 27,269 62,000	1,800 1,800 180 1,800	Static test to failure Static test to failure
D1B28 D1B34 D1B22 D1B47 D1B3	46 45 45 40 40	70,000 58,000 89,000 91,000 100,000	1,800 1,800 1,800 1,800 1,800	Failed 0.05 in. out of middle inch
D1B40 D1B35 D1B29 D1B33 D1B15	40 35 35 35 35 30	152,000 121,000 257,000 555,000 270,000	1,800 1,800 1,800 1,800 1,800	Failed 0.30 in. out of middle inch Failed 0.55 in. out of middle inch
D1B44 D1B38 D1B23 D1B9 D1B20	30 30 30 30 30 30	281,000 422,000 542,000 549,000 575,000	1,800 1,800 1,800 1,800 1,800	Failed 0.65 in. out of middle inch Failed 1.00 in. out of middle inch Failed 0.45 in. out of middle inch
D1B41 D1B26 D1B31 D1B13 D1B37	50 50 28 28 28 27	1,169,000 1,809,000 11,346,000 41,182,000 325,000	1,800 1,800 1,800 1,800 1,800	
D1B43 D1B27 D1B2 D1B21 D1B32 D1B14	27 25 25 25 25 25 25	1,331,000 403,000 1,064,000 88,719,000 89,122,000 92,802,000	1,800 1,800 1,800 1,800 1,800 1,800	Failed 0.76 in. out of middle inch Feiled 0.59 in. out of middle inch Did not fail

TABLE II.- FATIGUE TEST RESULTS FOR 61s-T6 ALLMINUM-ALLOY MATERIAL UNDER DIRECT STRESS AT R=0 - Continued

(b) Notched sheet specimens,  $K_T = 2$ 

Remurks		Speed, cpm	Fatigue life, cycles	Maximum stress, ksi	Specimen
test to failure				49.9	D1A6
test to failure				48.6	D1C1+7
test to failure	Static t		0 -	48.5	D1C41
		180	642	48.5	D1A42
		180	3,063	48.5	D1C34
		180	3,127	47	D1.A36
		180	4,682	47	D1A38
		180	6,550	45	D1A48
		180	7,195	45	D1A34
		180	10,988	42	D1A28
		1,800	7.000	40	D1A46
		1,800	8,000	40	D1A45
		1,800	12,000	40	D1A13
		1,800	23,000	35	D1C38
		1,800	25,000	35	D1A27
		1,800	30,000	35	D1C30
	u.	1,800	43,000	30	DLA3
		1,800	64,000	30	D1C42
		1,800	124,000	30	D1C12
		1,800	101,000	25	D1C24
		180	136,068	25	DLA10
		1,800	160,000	25	D1A15
		1,800	281,000	25	DLA31
		1,800	336,000	20	D1A25
		1,800	1,136,000	20	D1A33
		1,800	1,314,000	20	D1A21
surface flaw near notch	Failed at su	1,800	189,000	18	D1C36
		1,800	493,000	18	DIA18
		1,800	661,000	18	DIAL9
		1,800	764,000	18	D1A14
		1,800	721,000	16	DLA8
	1	1,800	10,487,000	16	DLA7
		1,800	12,010,000	16	D1A37
	Ì	1,800	23,008,000	14	DLAHI
	)	1,800	30,679,000	14	D1C32
	1	1,800	49,254,000	14	D1A26
Did not fail	Die	1,800	65,730,000	11.8	D1A20

TABLE II. - FATIGUE TEST RESULTS FOR 61S-T6 ALUMINUM-ALLOY MATERIAL

UNDER DIRECT STRESS AT R = 0 - Concluded

(c) Notched sheet specimens,  $K_T = 4$ 

Specimen	Maximum stress, ksi	Fatigue life, cycles	Speed, cpm	Remorks
D1C2	49.5			Static test to failure
D1C26	49.1			Static test to failure
D1C15	47	195	2	
D1C14	45	394	180	Ĺ
D1C19	45	690	180	
D1C29	42.5	388	180	
DlA9	42.5	525	180	į
Dla44	40	657	180	]
DICIL	40	688	180	
D1C27	35	1,410	180	
DLA4	35	2,235	180	
D1C33	30	2,700	180	
D1C22	30	3,735	180	1
D1C31	25	6,157	180	1
D1C37	25	6,489	180	
D1C3	25	6,765	180	
D1C45	20	20,000	1,800	ļ
D1C39	20	22,000	1,800	}
D1C5	15	74,271	180	}
D1A3	15	108,000	1,800	
D1C23	15	115,000	1,800	
D1A2	12	639,000	1,800	ì
DICL	10	424,000	1,800	
D1C35 -	10	429,000	1,800	1
D1C7	8	26,535,000	1,800	
D1C8	8	30,468,000	1,800	
D1C13	6	96,481,000	1,800	Did not fail
D1C25	6	103,261,000	1,800	Did not fail

### TABLE III. - FATIGUE TEST RESULTS FOR 347 STAINLESS-STEEL MATERIAL

### UNDER DIRECT STRESS AT R = 0

### (a) Unnotched sheet specimens

Specimen	Maximum stress, ksi	Fatigue life, cycles	Speed cpm	Remarks
E1B26	89.9			Static test to failure
ElC12	88	403	180	
ElC17	88	675	180	İ
ElB22	85	263	180	İ
E1B28	85	740	180	]
Enc16	85	9,363	180	
ELA28	85	12,540	180	}
E1B25	80	94,418	180	}
ElB17	80	98,000	1,800	Failed 0.75 in. out of middle inch
E1B24	75	49,340	180	
<b>E1</b> B16	73.4	45,000	1,800	
E1B12	73.2	72,000	1,800	†
ElBll	70	113,000	1,800	
ELB15	70	153,000	1,800	}
E1B7	65	168,000	1,800	
E1B5	65	204,000	1,800	
ELB23	60	206,000	1,800	
ELB27	60	233,000	1,800	
ElB18	57	478,000	1,800	
E1B19	57	697,000	1,800	Failed 1.40 in. out of middle inch
mB6	55	379,000	1.800	
E1B1	55	503,000	1,800	Failed 0.60 in. out of middle inch
ElB2	55	53,543,000	1,800	Did not fail
ELB9	53	59,529,000	1,800	Did not fail
ELB13	53	72,050,000	1,800	Did not fail
ELB10	50	30,261,000	1,800	Did not fail
ELB14	50	36,087,000	1,800	Did not fail

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TABLE III. - FATIGUE TEST RESULTS FOR 347 STAINLESS-STEEL MATERIAL

UNDER DIRECT STRESS AT R = 0 - Continued

(b) Notched sheet specimens,  $K_T = 2$ 

Specimen	Maximum stress, ksi	Fatigue life, cycles	Speed,	Remarks
ELA18 ELA12 ELD13 ELA17	97 95.2 94.5 90	1,219	2	Static test to failure Static test to failure Static test to failure
ELD9	85	82	180	
E1A22 E1A10 E1A16 E1A14 E1C14	85 85 80 80 75	2,603 5,195 6,091 9,263 9,651	180 180 180 180 180	
ELA4 ELC9 ELA8 ELA11 ELD8	75 -70 70 65 60	12,660 13,556 15,207 27,000 18,000	180 180 180 1,800	
EID16 EIC7 EIC3 EIC1 EIA25	60 55 55 50 50	35,000 50,000 54,000 84,000 94,236	1,800 1,800 1,800 1,800 180	
EID12 EIA6 EID24 EID20	50 45 .45 40 40	98,000 121,000 177,000 424,000 624,000	1,800 1,800 1,800 1,800 1,800	
E1C5 E1A3 E1A7 E1A26 E1C28	40 39 38 37 35	868,000 44,939,000 56,027,000 65,985,000 57,692,000	1,800 1,800 1,800 1,800 1,800	Did not fail Did not fail Did not fail Did not fail

TABLE III. - FATIGUE TEST RESULTS FOR 347 STAINLESS-STEEL MATERIAL

UNDER DIRECT STRESS AT R = 0 - Concluded

(c) Notched sheet specimens,  $K_{T} = 4$ 

Specimen	Maximum stress, ksi	Fatigue life, cycles	Speed,	Remarks
ELA15 ELC23 ELC8 ELC4 ELC27	85.5 85.25 83 83 80	36 52 100	2 2	Static test to failure Static test to failure
E1A24	80	378	180	
E1C24	70	848	180	
E1A23	70	1,375	180	
E1C26	65	1,689	180	
E1C11	65	1,910	180	
ELD17 ELC21 ELC18 ELC15 ELD5	60 60 50 50 50 40	2,245 3,014 11,000 12,000 17,367	180 180 1,800 1,800 180	
ELA5	40	40,000	1,800	
ELD21	40	48,000	1,800	
ELC6	30	160,000	1,800	
ELA1	30	217,000	1,800	
ELC25	28	214,000	1,800	
E1A9	28	20,054,000	1,800	Did not fail Did not fail Did not fail Did not fail Did not fail
E1C22	27	20,779,000	1,800	
E1C19	26	54,272,000	1,800	
E1A19	24	64,052,000	1,800	
E1A27	20	36,196,000	1,800	

TABLE IV. - FATIGUE TEST RESULTS FOR 403 STAINLESS-STEEL MATERIAL

### UNDER DIRECT STRESS AT R = 0

### (a) Unnotched sheet specimens

Specimen	Maximum stress, ksi	Fatigue life, cycles	Speed, cpm	Remurks
FLA30	194.6			Static test to failure
F1D1	185	15	130	
F1C19	182	9,720	180	
F1D19	182	9,915	180	
F2C2	180	4,710	180	
F1A2	<b>1</b> 50	8,418	180	
F1B14	170	14,619	180	
F1D18	160	31,000	1,800	
FLB5	160	<b>3</b> 8,000	1,800	
F1C15	מיו	51,000	1,800	
F1B19	150	52,000	1,800	
F1A28	140	48,000	1,800	Filled 0.20 in. out of middle inch
F1D20	140	64,000	1,800	
Flal9	140	72,000	1,800	
F1C8	130	74,000	1,800	
F1B28	130	97,000	1,800	
F1B2	120	85,000	1.800	Ì
F1C9	120	132,000	1,800	
F1B20	110	130,000	1,800	
FlCll	110	343,000	1,800	Failed 0.50 in. out of middle inch
F1C18	110	549,000	1.800	
F1A21	105	335,000	1,800	
F1C29	105	865,000	1,800	
F1A5	103	149,000	1,800	
FlA4	103	283,000	1,800	
F1B3	100	149,000	1,800	Failed at surface flaw
F1C5	100	420,000	1,800	
F1B16	100	33,871,000	1,800	Did not fail
F1B7	90	35,540,000	1,800	Did not fail

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TABLE IV. - FATIGUE TEST RESULTS FOR 403 STAINLESS-STEEL MATERIAL

UNDER DIRECT\_STRESS AT R = 0 - Concluded

(b) Notched sheet specimens,  $K_T = 2$ 

Specimen	Maximum stress, ksi	Fatigue life, cycles	Speed,		Remai	rks	
F1A33 F1C24 F1B4 F1B17 F1B31	207 195 175 150 1 <b>3</b> 5	716 834 3,232 6,648	2 2 180 180	Static	test	to	failure
F1B13 F1C23 F1D15 F1B1 F1B11	120 110 100 90 80	24,000 23,000 37,000 71,000 83,000	1,800 1,800 1,800 1,800				
F1B12 F1C3 F1A32 F1D5	70 65 62.5 60	170,000 284,000 218,000 22,236,000	1,800 1,800 1,800 1,800				

(c) Notched sheet specimens,  $K_{\rm T} = 4$ 

Specimen	Maximum stress, ksi	Fatigue life, cycles	Speed,	Remarks
F1A8 F1C6 F1C16 F1A25 F1D13	204 200 180 160 140	19 129 252 922	2 2 2 2 180	Static test to failure
F1A3 F1C28 F1C31 F1D8 F1A6 F1D14 F1D7	120 100 80 50 40 38 35	1,903 3,667 14,000 89,000 10,644,000 1,798,000 41,709,000	180 180 1,800 1,800 1,800 1,800 1,800	Did not fail

	l J	DIA13	DIAI9	DIA25	DIA3I	DIA37	DIA43
0187 DIBI3 DI		٥	DIBI9	DIB25	01831	D1837	DIB43
[ DICI3 ]		ā	DICI9	DIC25	DIC3I	DIC37	DIC43
		ā	120	DIA26	DIA32	DIA38	DIA44
DIBS DIBI4 DIE		Ö	DIB20	01826	DIB32	8£8IQ	DIB44
l Dici4		ìla	DIC20	DIC26	DIC32	DIC38	DIC44
I DIAIS		/ia	121	DIA27	DIA33	DIA39	DIA45
310 S1810 6810		310	01821	01827	E 8 8 10	62810	01845
do de la composição de la		ola	DIC21	DIC27	DIC33	DIC39	DIC 45
) DIAIG		71G	122	DIA28	DIA34	DIA40	DIA46
DIBIO DIBIG DIB22		<b>810</b>	22	D1828	01834	01840	01846
DIC 1 DICIE 1 DIC		)IO	DIC22	DIC28	DIC34	DIC40	DIC 46
DIAII I DIAI7 I DIA		7IQ	123	DIA29	DIA 35	DIA41	DIA47
DIBIT DIBIT DIB			DIB23	DIB 29	<b>95 810</b>	01841	01847
I DICI7 I	DICIZ   DIC	) П	DIC23	DIC29	DIC35	DIC41	DIC 47
I DIAI8 I		DIA	24	DIA30	DIA36	DIA42	DIA48
DIBI2 DIBI8 DIB		DIE	<b>JIB24</b>	DIB30	988IQ	D1842	01848
DIC 12   DIC 18   DI		ā	JIC24	DIC30	DIC36	DIC42	DIC48

Figure 1.- Sheet layout for 61S-T6 aluminum-alloy material. Sheet size, 4 feet by 12 feet.

<u>.</u>

FID28	EID24	EID20	EID16	EIDI 2	EID8	EID4
EIC28	EIC24	EIC20	EICI6	EICI 2	EIC8	EIC4
EIB28	E1B24	EIB20	EIBI6	EIBI 2	EIB8	E184
EIA28	E1A24	EIA20	EIAI6	EIAI 2	EIA8	EIA4
EIC27	E1C23	EICI9	EICI5	EICH	EIC 7	EIC3
E1827	E1B23	EIBI3	EIBIS	EIBII	E18.7	E1B3
EIA27	EIA23	EIAI9	EIAIS	EIAI I	EIA 7	EIA3
EIA26	EIA22	EIAI8	EIA14	EIAIO	EIA6	EIA2
EIB26	E1B22	EIB18	EIBI4	EIBIO	EIB6	E1B2
EIC26	EIC22	EIC 18	EIC14	ECIO	EICE	EIC2
EIA25	EIA21	EIA 17	EIA13	EM9	EIA5	EIAI
E1B25	EIB21	EIB17	EIBI3	EB3	E1B5	EIBI
EIC25	EIC21	EIC17	EICI3	EC9	EIC5	EICI
EID25	EID21	EIDI7	EIDI3	EID9	EID5	EIDI

Figure 2.- Sheet layout for 347 stainless-steel material. Sheet size, 3 feet by 10 feet.

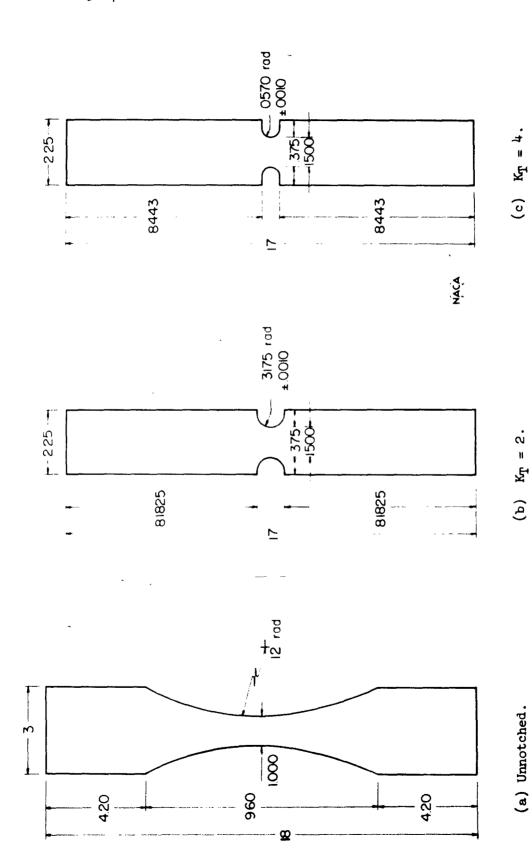


Figure 3.- Specimen configurations. (All dimensions are in inches.)

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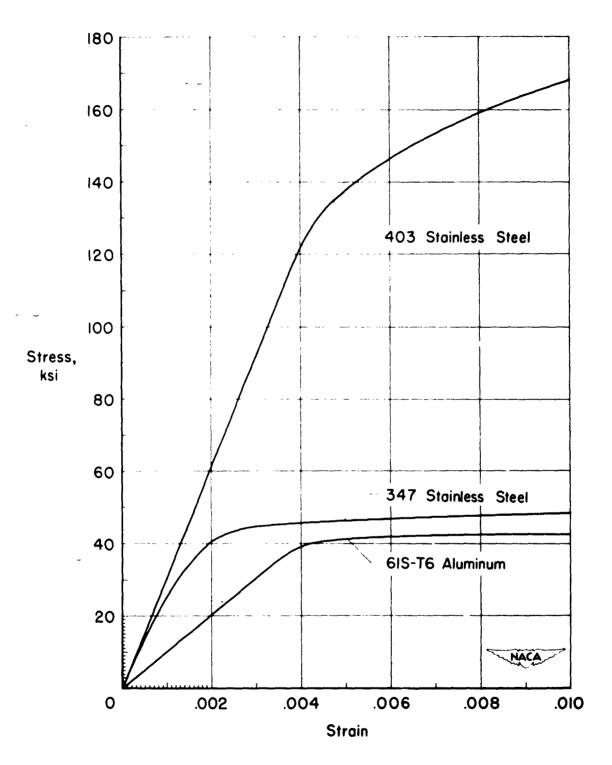


Figure 4.- Tensile stress-strain curves for materials used.

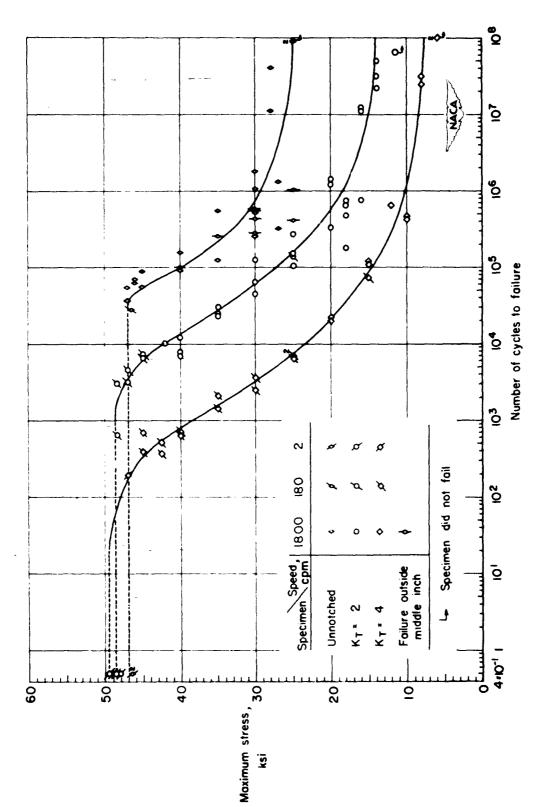


Figure 5.- Results of fatigue tests on 61S-T6 aluminum-alloy specimens under axial load at R=0.

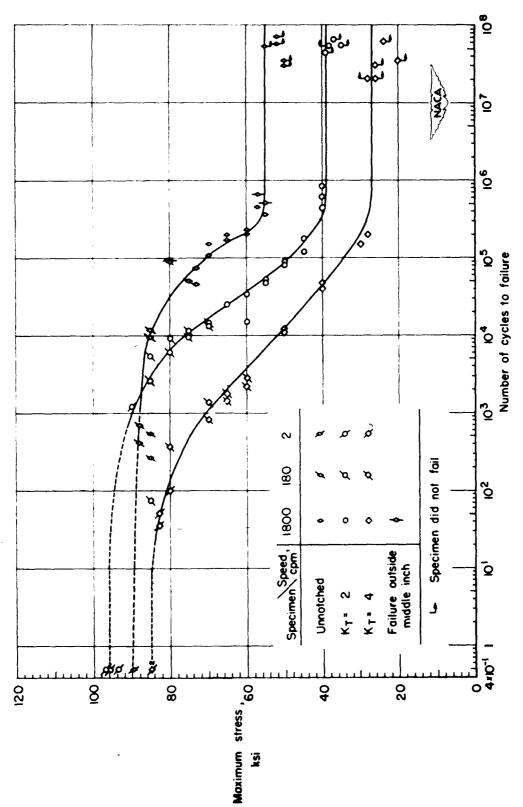


Figure 6.- Results of fatigue tests on 347 stainless-steel specimens under axial load at R=0.

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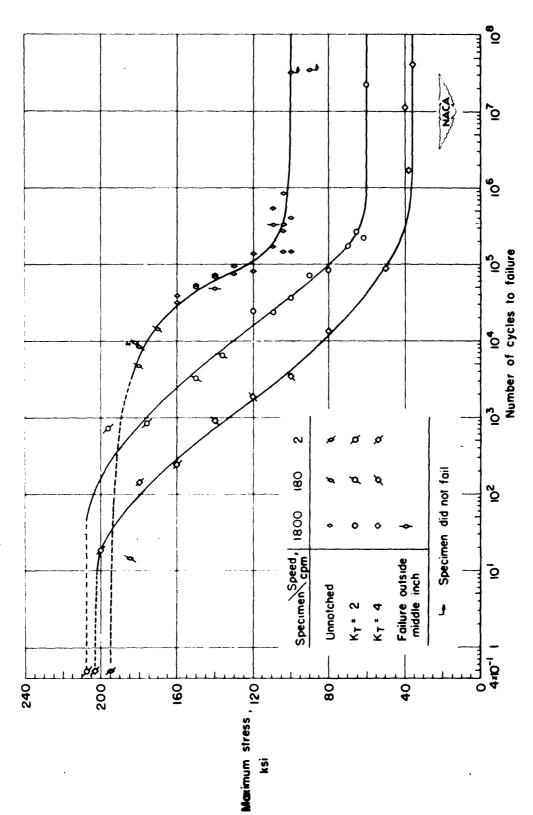


Figure 7.- Results of fatigue tests on 403 stainless-steel specimens under axial load at R = 0.

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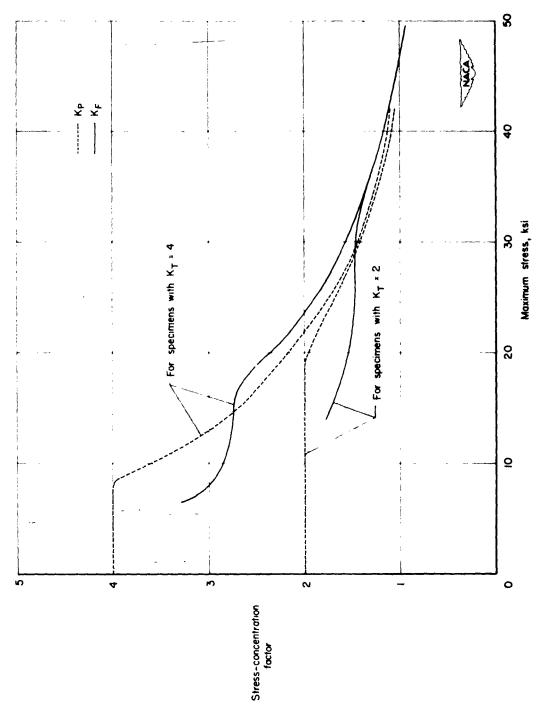
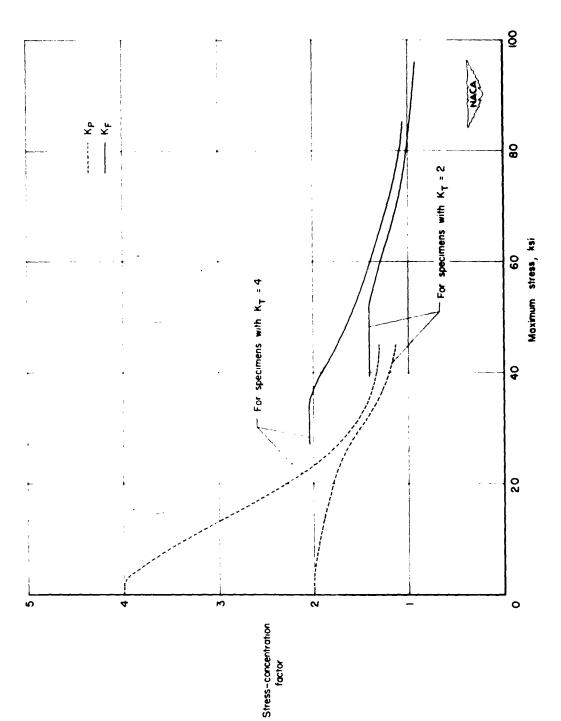
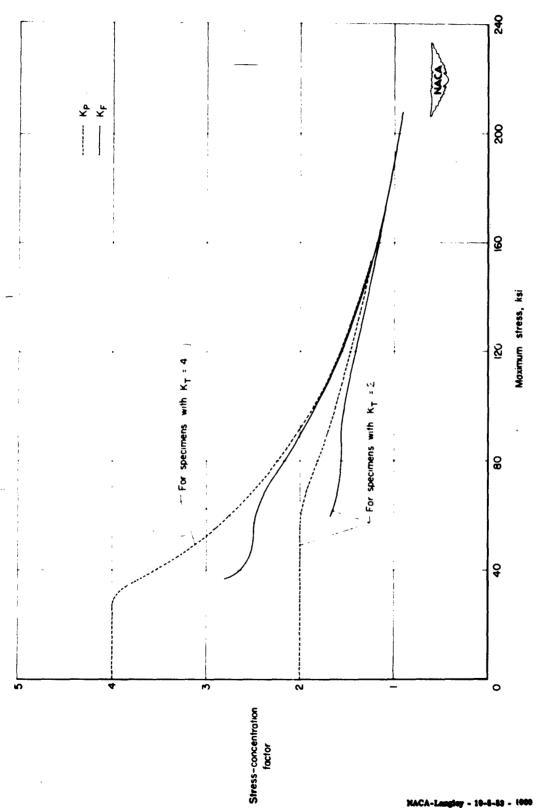


Figure 5.- Variation in  $K_{F}$  for tests on 618-76 aluminum-alloy specimens.



for tests on 347 stainless-steel specimens. Figure 9.- Variation in Kg

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for tests on 403 stainless-steel specimens. Figure 10.- Variation in KF

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FEEL. Herbert F. Hardrath, Charles B. Landers and Elmer C. Utley, Jr. October 1953, 28p. AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND UNNOTCHED SHEET SPECIMENS OF 61S-T6 ALUMINUM ALLOY, ANNEALED 347 STAINLESS STEEL, AND HEAT-TREATED 403 STAINLESS National Advisory Committee for Aeronautics. Magre., 4 tabe. (NACA TN 3017) **KACA** TN 3017

epecimens of 618-T6 aluminum alloy and 347 and 668 stainless steels. Special emphasis was placed theoretical elactic values at low stresses and were approximately equal to one at the ultimate strength. The salmimum life to failure at stresses near the on tests at high stress levels producing failures in tress-concentration factors effective in fatigue of ctched specimens were somewhat less than the Azial-load fatigue tests at a stress ratio of zero **Pertormed** on notched and unnotched sheet

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ALTMEINUM ALLOY, ANNEALED 347 STAINLESS
STEEL, AND HEAT-TREATED 403 STAINLESS
STEEL, Herbert F. Hardrath, Charles B. Landers
and Elmer C. Ulley, Jr. October 1953, 28p, AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND **Rational Advisory Committee for Aeronautics.** (NACA TN 3017) Megra, 4 tabe. GCA TN 3017

600 stainless steels. Special emphasis was placed interest specimens were somewhat less than the theoretical elastic values at low stresses and were continuedly equal to one at the ultimate strength. to beats at high stress levels producing failures in ess-concentration factors effective in fatigue of pecimens of 618-T6 aluminum alloy and 347 and **Recommended equa**lity was an use animals are used. Animi-load intigue tests at a stress ratio of zero mall members of cycles. It was found that the Here performed on notched and unnotched sheet

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(5.2.5)- Planticity (5. 2. 13) Materials, Properties - Fatigue

Hardrath, Herbert F. Utley, Elmer C., Jr. Landers, Charles B.

NACA TN 3017

Concentrated (4.3.7.6) Loads and Stresses, Loads and Stresses, Structural -

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STEEL. Herbert F. Hardrath, Charles B. Landers AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND ALUMINUM ALLOY. ANNEALED 347 STAINLESS STEEL, AND HEAT-TREATED 403 STAINLESS National Advisory Committee for Aeronautics. UNNOTCHED SHEET SPECIMENS OF 61S-T6 and Elmer C. Utley, Jr October 1953. 28p. diagrs., 4 tabs. (NACA TN 3017) PACA TR 3017

103 stainless steels. Special emphasis was placed approximately equal to one at the ultimate strength. on tests at high stress levels producing failures in theoretical elastic values at low stresses and were stress-concentration factors effective in fatigue of specimens of 618-T6 aluminum alloy and 347 and The mimimum life to failure at stresses near the Axial-load fatigue tests at a stress ratio of zero notched specimens were somewhat less than the small numbers of cycles. It was found that the were performed on notched and unnotched sheet

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ALUMINUM ALLOY, ANNEALED 347 STAINLESS STEEL, AND HEAT-TREATED 403 STAINLESS STEEL. Herbert F. Hardrath, Charles B. Landers National Advisory Committee for Aeronautics.
AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND UNNOTCHED SHEET SPECIMENS OF 618-T6 and Elmer C. Utley, Jr. October 1953. diagrs., 4 tabs. (NACA TN 3017) NACA TN 3017

403 stainless steels. Special emphasis was placed approximately equal to one at the ultimate strength. theoretical elastic values at low stresses and were on tests at high stress levels producing failures in stress-concentration factors effective in fatigue of specimens of 618-T6 aluminum alloy and 347 and The mimimum life to failure at stresses near the Axial-load fatigue tests at a stress ratio of zero notched specimens were somewhat less than the small numbers of cycles. It was found that the were performed on notched and unnotched sheet

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STEEL. Herbert F. Hardrath, Charles B. Landers AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND UNNOTCHED SHEET SPECIMENS OF 615-T6 ALUMINUM ALLOY, ANNEALED 347 STAINLESS STEEL, AND HEAT-TREATED 403 STAINLESS National Advisory Committee for Aeronautics. and Elmer C. Utley, Jr. October 1953. 28p. diagra., 4 tabs. (NACA TN 3017) VACA TR 3017

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STERIL. Herbert F. Hardrath, Charles B. Landers and Elmer C. Utley, Jr. October 1953. 28p. National Advisory Committee for Aeronautica.

ANIAL-LOAD FATIGUE TESTS ON NOTCHED AND ALUMINUM ALLOY, ANNEALED 347 STAINLESS STEEL, AND HEAT-TREATED 403 STAINLESS UNICHED SHEET SPECIMENS OF 618-T6 dagra., 4 tabs. (NACA TN 3017)

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Utley, Elmer C., Jr. NACA TN 3017

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Herbert F. Hardrath, Charles B. Landers AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND ALUMINUM ALLOY. ANNEALED 347 STAINLESS STEEL, AND HEAT-TREATED 403 STAINLESS National Advisory Committee for Aeronautics. UNNOTCHED SHEET SPECIMENS OF 61S-T6 and Elmer C. Utley, Jr October 1953. 28p. diagre., 4 tabs. (NACA TN 3017) STEEL.

Loads and Stresses,

403 stainless steels. Special emphasis was placed theoretical elastic values at low stresses and were approximately equal to one at the ultimate strength. on tests at high stress levels producing failures in stress-concentration factors effective in fatigue of specimens of 618-T6 aluminum alloy and 347 and The mimimum life to failure at stresses near the Axial-load fatigue tests at a stress ratio of zero notched specimens were somewhat less than the small numbers of cycles. It was found that the were performed on notched and unnotched sheet

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Utley, Elmer C., Jr.

NACA TN 3017

Landers, Charles B.

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STEEL. Herbert F. Hardrath, Chinies P. Landers AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND ALUMINUM ALLOY. ANNEALED 347 CTAINLESS **Stational Advisory** Committee for Aeronautics. UNNOTCHED SHEET SPECIMENS OF 61S-T6 md Elmer C. Utley, Jr. October 1953. 28p.

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National Advisory Committee for Aeronautics. **NACA TN 3017** 

AXIAL-LOAD FATIGUE TESTS ON NOTCHED AND UNNOTCHED SHEET SPECIMENS OF 61S-T6

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Loads and Stresses,

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(4. 3. 7. 7. 1)

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Seels

Structural - Repeated

Loads and Stresses,

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- Tensile - Fatigue

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Materials, Properties Materials, Properties Materials, Properties

STEEL, AND HEAT-TREATED 403 STAINLESS STEEL. Herbert F. Hardrath, Charles B. Landers and Elmer C. Utley, Jr. October 1953. 28p. ALUMINUM ALLOY, ANNEALED 347 STAINLESS

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(5.1.1)

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Axial-load fatigue tests at a stress ratio of zero were performed on notched and unnotched sheet diagrs., 4 tabs. (NACA TN 3017)

403 stainless steels. Special emphasis was placed on tests at high stress levels producing failures in specimens of 618-T6 aluminum alloy and 347 and small numbers of cycles. It was found that the

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Hardrath, Herbert F.

Utley, Elmer C., Jr. Landers, Charles B.

**NACA TN 3017** 

Hardrath, Herbert F. Utley, Elmer C., Jr.

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